

# Computational investigations of mechanical and dynamical properties of gold-based compounds ( $X_3\text{Au}$ , $X = \text{Ti}$ , $\text{Zr}$ and $\text{V}$ )



Selgin Al<sup>a,\*</sup>, Nihat Arikan<sup>b</sup>, Mustafa Özduran<sup>a</sup>, Ahmet İyigör<sup>c</sup>

<sup>a</sup> Ahi Evran University, Faculty of Science and Arts, Physics, Kırşehir 40100, Turkey

<sup>b</sup> Ahi Evran University, Education Faculty, Math and Science Department, Kırşehir 40100, Turkey

<sup>c</sup> Ahi Evran University, Central Research Lab, Kırşehir 40100, Turkey

## ARTICLE INFO

### Keywords:

DFT  
Direct-method  
Band structure  
Elastic constants  
Phonon

## ABSTRACT

The structural, elastic, electronic and phonon properties of  $X_3\text{Au}$  ( $X = \text{Ti}$ ,  $\text{Zr}$  and  $\text{V}$ ) compounds in the A15 structure were obtained in the framework of the density functional theory (DFT) within the generalized gradient approximation (GGA). The equilibrium lattice constants, bulk modulus and elastic constants were calculated. The calculation of elastic constants revealed that  $\text{V}_3\text{Au}$  has the highest hardness nature and incompressibility along the x-axis among them. The computed elastic constants also provided information about the ductility of  $X_3\text{Au}$  compounds which were predicted using Pugh's criteria. The results indicated that all three compounds have ductile nature. The density of states calculations revealed that electrons of Ti, Zr and V provide most contribution to the conductivity of the compounds and thus cause a metallic bonding. The investigation of stability via phonon spectra of compounds showed that these compounds are dynamically stable in the A15 structure.

## 1. Introduction

Intermetallic compounds with the A15 type structure (space group  $\text{Pm} - 3\text{n}$ ) and  $\text{A}_3\text{B}$  (A is a transition metal and B can be any element) composition mostly exhibit excellent properties such as high superconducting transition temperatures, melting points and critical field [1,2]. Therefore, there are ongoing investigations and renewed interest on those materials, especially with those having high superconducting transition temperatures. A number of experimental and theoretical studies have been carried out in order to understand structural, elastic, electronic, phase stability and thermodynamic properties for  $X_3\text{Au}$  ( $X = \text{Ti}$ ,  $\text{Zr}$  and  $\text{V}$ ) in the A15 structure [1,3–27]. The structural, elastic and electronic properties of  $\text{Ti}_3\text{Au}$  have been investigated theoretically using density functional theory [1,5]. Shinohara et al. [3] have measured the effect of heat treatment temperature on the precipitation behaviour of A15  $\text{Ti}_3\text{Au}$  using by tensile tests, X-ray diffraction measurement (XRD) and scanning electron microscopy (SEM) observation. The diffusion parameters and activation energies were obtained for  $\text{Ti}_3\text{B}$  ( $\text{B} = \text{Au}$ ,  $\text{Pt}$ ,  $\text{Ir}$ ) [6].  $\text{Zr}_3\text{Au}$  has been studied very earliest in terms of its structural, thermodynamic and formation enthalpy using XRD [12–16]. Unfortunately, the physical properties of this compound studied less in literature. Recently, the electronic, structural and thermodynamic properties of vanadium intermetallic compounds  $\text{V}_3\text{B}$  ( $\text{B} = \text{Au}$ ,  $\text{Pt}$ ,  $\text{Ir}$ ) have been explored using first principle calculations by Wang et al. [21]. The effect of crystallographic structure on superconducting transition temperature was investigated by Blaughner et al. [22] for a number of intermetallic compounds. The tracer diffusion coefficients of vanadium, gallium and silicon for  $\text{V}_3\text{Ga}$  [23] and  $\text{V}_3\text{Si}$  [24] were determined using the tracer and diffusion coupling methods.  $\text{V}_3\text{Au}$  demonstrated increasing transition temperature (from 0.980 K to 1.7850 K) after annealing at high

\* Corresponding author.

E-mail address: [selgin.al@ahievran.edu.tr](mailto:selgin.al@ahievran.edu.tr) (S. Al).

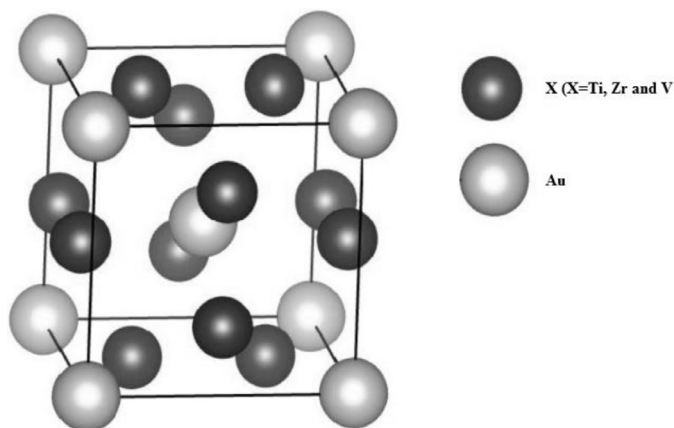


Fig. 1. Crystal structure of  $X_3Au$  compounds in the A15 structure.

**Table 1**

Calculated lattice constant  $a$ , bulk modulus  $B$  (GPa), elastic constants  $C_{ij}$  (GPa) and shear modulus  $G$  (GPa),  $B/G$  for  $X_3Au$  ( $X = Ti, Zr$  and  $V$ ) compounds, compared with the available experimental and theoretical data.

Materials		$a$	$B$	$C_{11}$	$C_{12}$	$C_{44}$	$G$	$B/G$
$Ti_3Au$	This work	5.104	135.086	245.086	80.086	58.984	67.483	2.002
	TB-LMTO [1]	5.0875	137					
	FP-LAPW [5]	5.112	140.162	287.776	71.461	70.311	83.502	1.678
	Exp. [10]	5.098						
	Exp. [20]	5.0960						
	Theory [11]	5.1						
	Theory [17]	5.098						
$Zr_3Au$	Exp. [18]	5.0974						
	Exp. [19]	5.0974						
	This work	5.518	116.758	213.591	68.342	37.417	48.959	2.385
$V_3Au$	Exp. [14]	5.483						
	Theory [17]	5.486						
$V_3Au$	This work	4.867	193.113	359.695	53.473	111.022	75.541	2.567
	Exp. [9]	4.874						
	Exp. [14]	4.883						
	Theory [17]	4.884						
	Exp. [18]	4.8813						
	Exp. [19]	4.881						
	Exp. [25]	4.882						
	Exp. [20]	4.87						
	CASTEP-GGA[21]	4.887	189.7	339.6	45.0	114.8	65.6	2.894
	CASTEP-LDA [21]	4.784	225.6	387.6	48.0	144.6	70.3	3.208

temperature in their results. The transition temperature of  $V_3Au$  from electronic structure results was obtained by Jarlborg et al. [26]. The impact of high pressure on transition temperature for  $V_3Au$  was explored by Smith [27]. Even though, several structural and thermodynamic data exist in literature for  $V_3Au$  and  $Ti_3Au$  compounds, there is less study about elastic properties and electronic band structure of  $Zr_3Au$  and all compounds' phonon properties which are crucial in order to understand lattice dynamics in microscopic scale. This kind of knowledge can provide information about phase transition, thermodynamic stability, defect transport and thermal properties. Thus, this study considers whole investigations about the following three A15 type compounds, namely;  $Ti_3Au$ ,  $Zr_3Au$  and  $V_3Au$  in terms of structural, elastic properties, electronic and phonon using first principles calculations. Our first principles calculations also provide information about chemical bonding, ductile and electronic density of states of the compounds that allow us to compare the data with the existing as well as adding new data to the literature.

## 2. Method of computation

The calculations were carried out by using MedeA-VASP package [28,29] using projected augmented wave method [30]. This is based on first principles calculations within the plane-wave pseudopotential density functional theory (DFT) method. The exchange-correlation energy was calculated by using the generalized gradient approximation of Perdew–Burke–Ernzerhof (PBE) [31]. A cut off energy of 500 eV was used in all calculations of structural, elastic and phonon properties. A  $10^{-9}$  eV energy convergence criteria was used within Normal (blocked Davidson) and reciprocal space projection operators. An  $8 \times 8 \times 8$  k-point mesh for all  $X_3Au$  ( $A = V, Ti$  and  $Zr$ ) compounds following the convention of Monkhorst and Pack [32] was conducted for the best convergence and relaxation

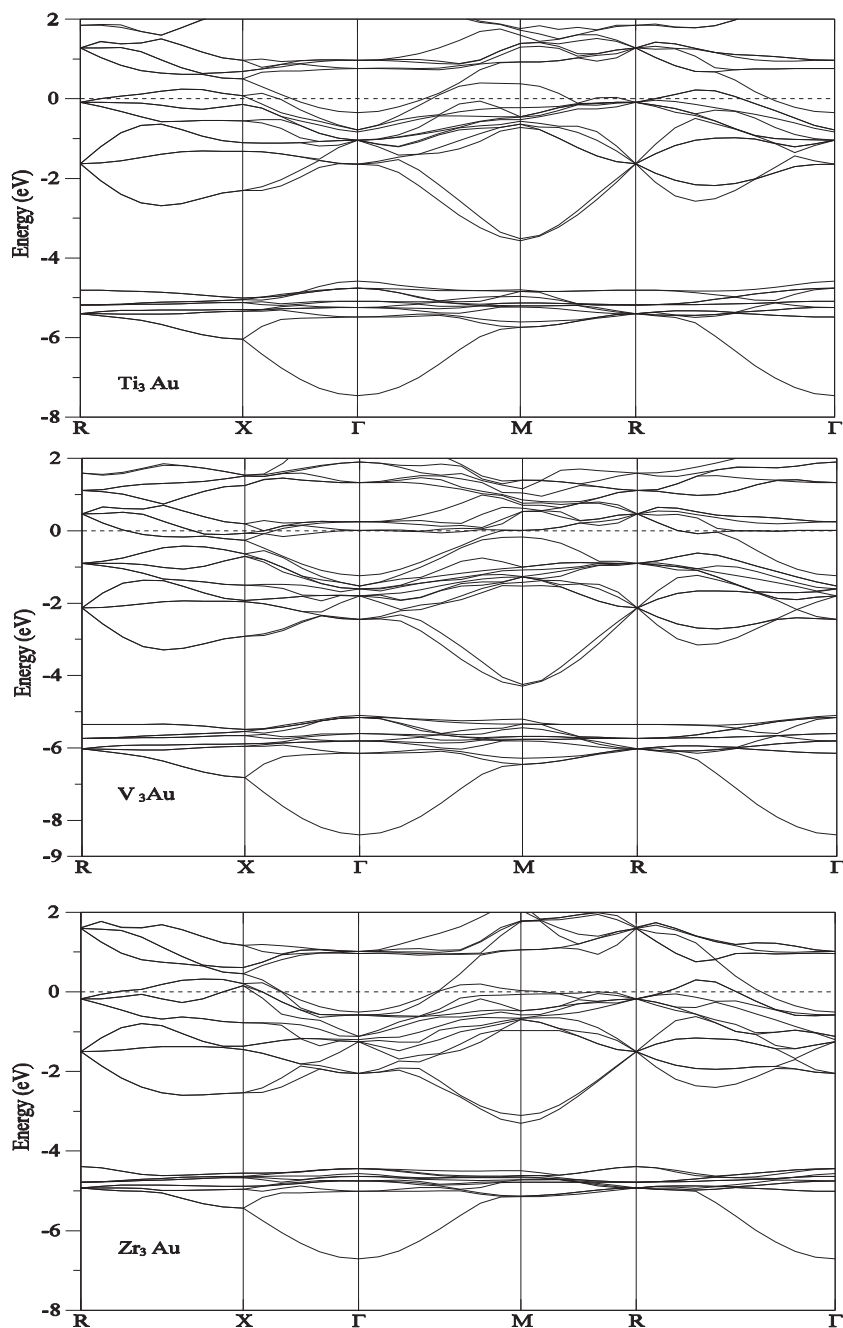


Fig. 2. The electronic band structure of  $X_3\text{Au}$  ( $X = \text{Ti}, \text{Zr}$  and  $\text{V}$ ) in the A15 structure.

zero strain. The Methfessel-Paxton type was used to smear Fermi level [33] with a width of 0.225 eV. The elastic constants were calculated using stress-strain approach [34], whereas the stress due to an applied strain was directly computed. The phonon spectra were obtained using the MedeA-Phonon together with MedeA-Vasp by the direct method. The simulation cell consisted of 54 atoms with periodic boundary conditions  $3 \times 3 \times 3$  supercells.

### 3. Result

The study of gold-based compounds  $X_3\text{Au}$  ( $X = \text{Ti}, \text{Zr}$  and  $\text{V}$ ) is an important step to understand the physical properties of intermetallic compounds containing rare earth elements. It was said that  $X_3\text{Au}$  ( $X = \text{Ti}, \text{Zr}$  and  $\text{V}$ ) compounds are stable in A15 structure [14]. Those are in the space group  $\text{Pm-}3\text{n}$ , with Au and X atoms located at Wyckoff position 2a (0, 0, 0) and 6c (1/4, 0, 1/2), respectively (see Fig. 1). For all the investigated compounds in this study, the calculated equilibrium lattice constants and the bulk

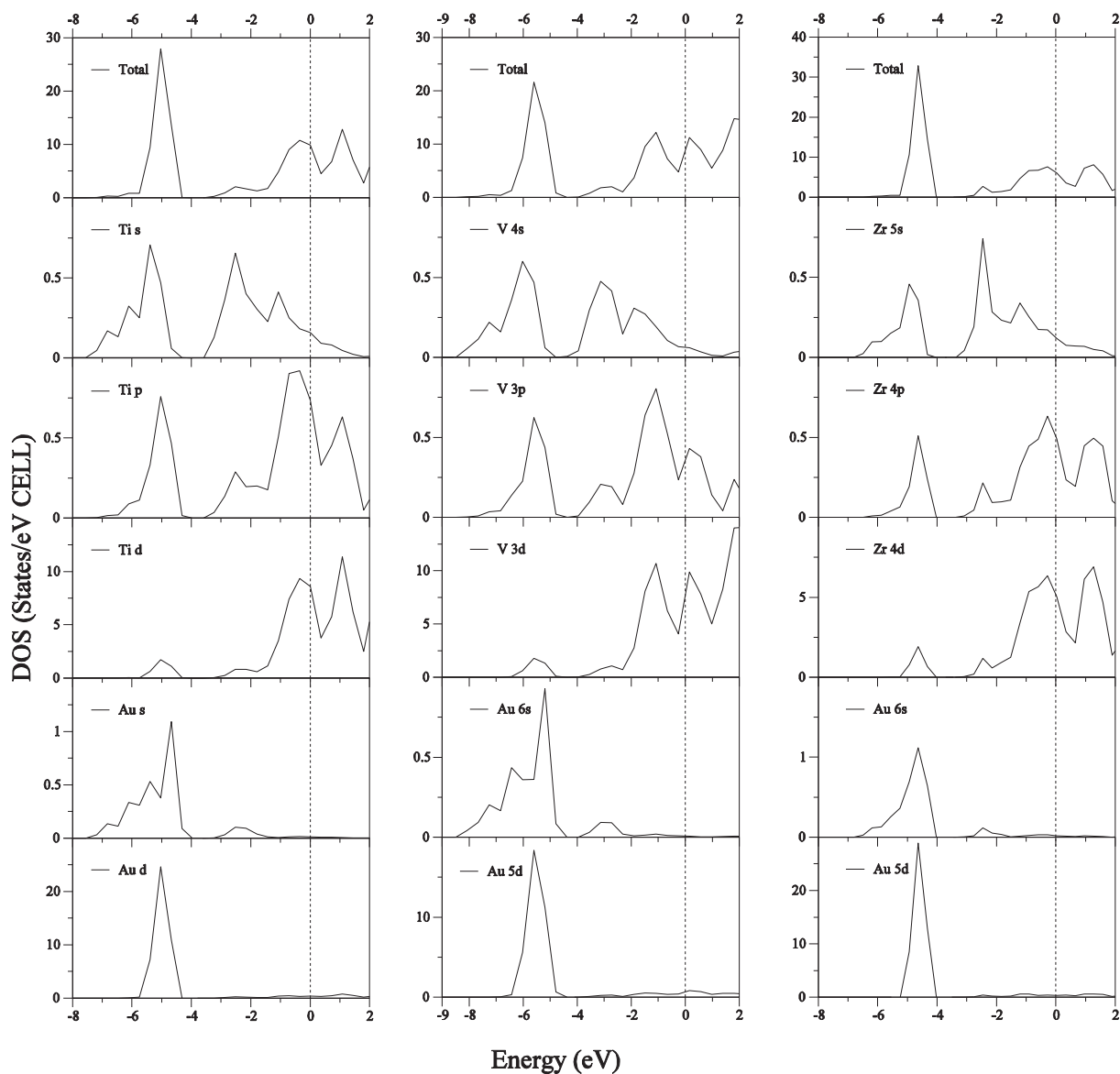


Fig. 3. The total and partial density of states for  $X_3\text{Au}$  ( $X = \text{Ti}, \text{Zr}$  and  $\text{V}$ ) in the A15 structure.

modulus are presented in Table 1 together with the available experimental and theoretical data. The computed values of the lattice constants for  $\text{Ti}_3\text{Au}$ ,  $\text{Zr}_3\text{Au}$  and  $\text{V}_3\text{Au}$  are fairly in a good agreement with the given data in Table 1, the deviations are only about 0.1%, 0.6% and 0.3%, respectively. Our computed values of bulk modules for  $\text{Ti}_3\text{Au}$  and  $\text{V}_3\text{Au}$  are 135.13 GPa and 193.113 GPa, respectively. The obtained bulk module for  $\text{Ti}_3\text{Au}$  is less than the values that were calculated by Sathyakumari et al. [1] and Rajagopalan et al. [5]. The reason for this might be the difference in lattice constants. If the lattice constant is higher as in ours, the bulk module tends to be smaller. On the other hand, our computed bulk module of  $\text{V}_3\text{Au}$  is higher than the one that was obtained by Wang et al. [21]. This is also due to the difference in lattice constants. Unfortunately, there is no reported data for bulk module of  $\text{Zr}_3\text{Au}$  in literature for comparison. This is reported in this work for the first time to best of our knowledge. The overall bulk modules of  $\text{Ti}_3\text{Au}$  and  $\text{V}_3\text{Au}$  are in a good agreement with the previous values [1,5,21].

Elastic constants describe the mechanical resistance of crystal materials against external applied stress. It is enough to define elastic properties for a cubic crystal with three independent elastic constants; namely  $C_{11}$ ,  $C_{12}$  and  $C_{44}$ . The computed values of three elastic constants for these materials are presented in Table 1, along with the theoretical results for comparison. Our computed values of the elastic constants of all compounds satisfy Born's [35] stability conditions for cubic crystals.

$$C_{11} + 2C_{12} > 0; C_{11} - C_{12} > 0; C_{44} > 0 \quad (1)$$

The calculated elastic constants for  $\text{Ti}_3\text{Au}$  and  $\text{V}_3\text{Au}$  are in a good accordance with the theoretical results [5,21]. As it is clear from

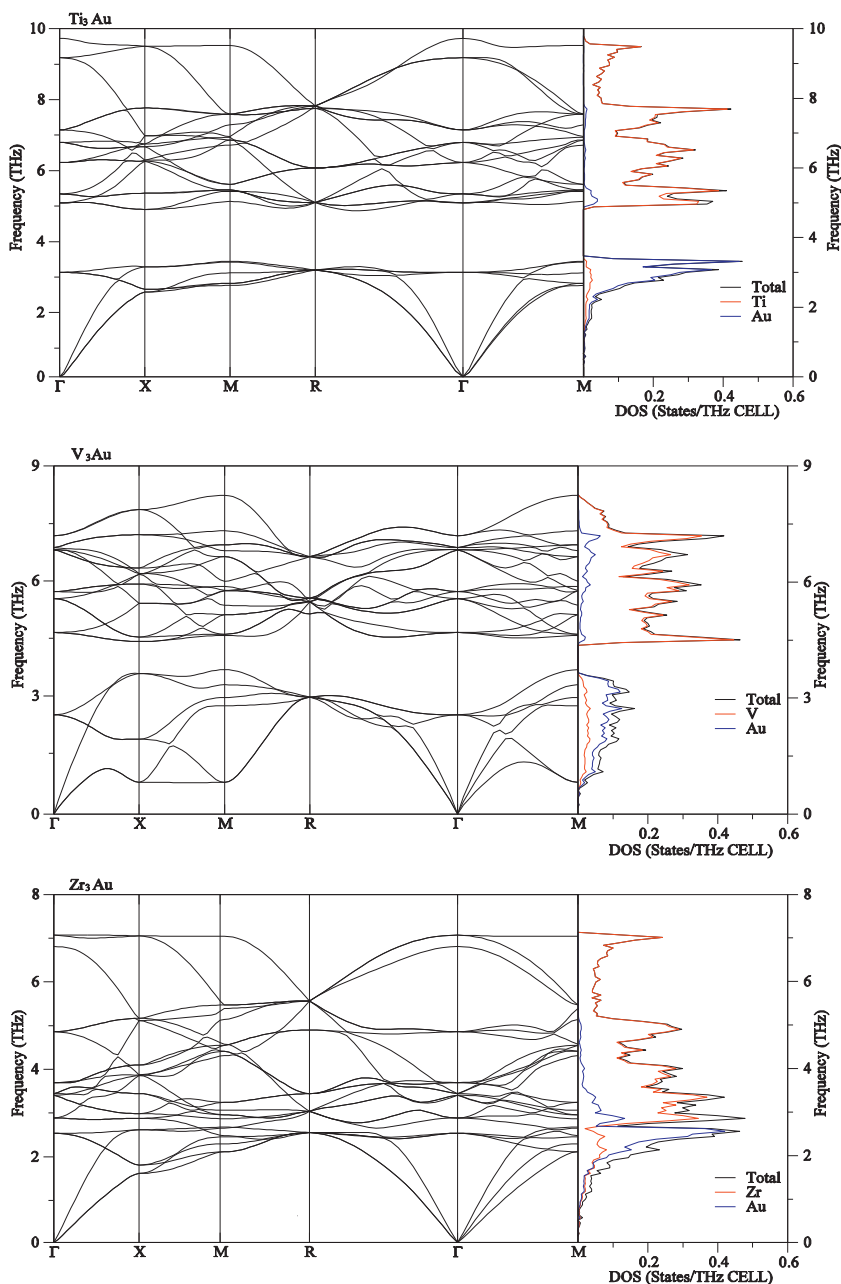


Fig. 4. The phonon dispersion curves, total and projected density of states for  $X_3\text{Au}$  ( $X = \text{Ti}, \text{Zr}$  and  $\text{V}$ ) in the A15 structure.

Table 1 that the value of  $C_{11}$  is higher among the calculated elastic constants for all three compounds. The importance of the value of  $C_{11}$  is that it provides an insight about compressibility along the x-axis. Higher value of  $C_{11}$  indicates that  $X_3\text{Au}$  compounds are incompressible under applied stress along the x-axis.  $\text{V}_3\text{Au}$  seems to be least compressive material due to having the highest  $C_{11}$  value among the three compounds. In addition, the hardness of the material can be determined by evaluating  $C_{44}$  value. The higher value of  $C_{44}$  shows the ability to resist monoclinic shear in the (001) plane and so thus the highest hardness. Hence,  $\text{V}_3\text{Au}$  exhibits the highest hardness among these compounds. The hardness of a material can also be determined by evaluating bulk modulus indirectly. From Table 1, it is clearly seen that the bulk modulus of  $\text{V}_3\text{Au}$  is the highest among them. This suggests the highest hardness of this material.

On the other hand, the  $B/G$  ratio provides quantitative information about the brittle or ductile nature of the compounds, as proposed by Pugh [36]. If the  $B/G$  ratio is greater than 1.75, the compound has a ductile nature. At the opposite case, the compound is brittle. Our calculated  $B/G$  values for  $\text{Ti}_3\text{Au}$ ,  $\text{Zr}_3\text{Au}$  and  $\text{V}_3\text{Au}$  are 2.002, 2.385 and 2.567, respectively. It is clear that all three compounds exhibit a ductile property which contradicts the result reported by Rajagopalan et al. [5].

The density of states and electronic band structures give information about bonding characteristics of compounds and their mechanical stability. Electrons with energies at the Fermi level play an important role in chemical bonding. The electronic band structures for  $X_3\text{Au}$  ( $X = \text{Ti, Zr and V}$ ) are presented in Fig. 2, in which  $E_F = 0$ . These materials show metallic behaviour since there is no energy gap at the Fermi level in Fig. 2. The calculated band structures of  $\text{Ti}_3\text{Au}$  and  $\text{V}_3\text{Au}$  are in well accordance with previously reported data [1,5,26]. Again, there is no existing data for  $\text{Zr}_3\text{Au}$  for comparison. The electronic structure of  $\text{Zr}_3\text{Au}$  is firstly reported in this study. It is seen that the electronic band structures of all three compounds are similar to each other. The total and projected density of states (DOS) to understand the band characteristics of all three compounds are presented in Fig. 3. The main contributions to the density of states at the Fermi level for  $\text{Ti}_3\text{Au}$ ,  $\text{Zr}_3\text{Au}$  and  $\text{V}_3\text{Au}$  are Ti d state, Zr d states and V d states, respectively. These results indicate that electrons of Ti, Zr and V provide most contribution to the conductivity of the compounds and thus cause a metallic bonding. The partial density of states for Ti and Au exhibited in Fig. 3 that s- and p- electrons of Ti are dominant in the region of  $E < E_F$  which suggests that the bonding of  $\text{Ti}_3\text{Au}$  is determined by s- and p- electrons of Ti. Similar effects are seen in partial density of states for  $\text{Zr}_3\text{Au}$  and  $\text{V}_3\text{Au}$  with participation of d- electrons. The results also display a sharp peak below  $-4\text{ eV}$  for all three compounds which are caused by Au d states.

The stabilities of compounds are associated with low value of density of state (DOS) at the Fermi energy  $E_F$  [37]. A lower value of  $N(E_F)$  is an indication of a higher stable phase. For all three compounds in A15 structure, the total state density at the Fermi level  $N(E_F)$  was calculated. This calculated values of this parameter are 9.8, 9.1 and 7.2 states/cell.eV for  $\text{Ti}_3\text{Au}$ ,  $\text{Zr}_3\text{Au}$  and  $\text{V}_3\text{Au}$ , respectively. According to these  $N(E_F)$  values,  $\text{Zr}_3\text{Au}$  has better structural stability than the other two compounds. As can be seen from Fig. 3 that Ti d, Zr d and V d states in  $X_3\text{Au}$  ( $X = \text{Ti, Zr and V}$ ) shift to higher energies in the following sequence from  $\text{Zr}_3\text{Au}$ , to  $\text{V}_3\text{Au}$  and to  $\text{Ti}_3\text{Au}$ .

The per primitive cubic unit cell of a A15 structure contains eight atoms, corresponding to twenty four phonon modes in the dispersion relations, with three of them being the acoustic mode. The calculated full phonon spectra and their projected density of states for  $X_3\text{Au}$  compounds are displayed in Fig. 4. These compounds are dynamically stable since the phonon frequencies  $\omega$  in all directions of the  $X_3\text{Au}$  compounds are not imagers. For  $\text{Ti}_3\text{Au}$  and  $\text{V}_3\text{Au}$ , the acoustic and the lower optical modes overlap and the other upper optical modes are separated from the rest of the phonon modes. These gaps of  $\text{Ti}_3\text{Au}$  and  $\text{V}_3\text{Au}$  compounds are calculated as 1.5 THz and 2.1 THz, respectively. For  $\text{Ti}_3\text{Au}$  and  $\text{V}_3\text{Au}$ , the phonon frequencies above the band gap are completely composed of the vibrations of the Ti and V atoms, as the atomic mass of Au is heavier than the atomic mass of them. For  $\text{Zr}_3\text{Au}$ , the upper optical region entirely consists of the vibrations of Zr atoms, while and the acoustic and the lower optical region is due to the vibrations of Au atoms. Unfortunately, theoretical or experimental phonon spectrums of  $X_3\text{Au}$  ( $X = \text{Ti, Zr and V}$ ) compounds are not available in literature for further comparison. Our phonon calculations of these compounds are in a good agreement with our earlier work [38,39] which belongs to the some space groups.

#### 4. Conclusions

In this study, the structural, elastic, electronic and phonon properties of  $X_3\text{Au}$  ( $X = \text{Ti, Zr and V}$ ) compounds in the A15 structure were investigated in the framework of the density functional theory (DFT) within the generalized gradient approximation (GGA). The structural properties such as the lattice constants and bulk modules were computed and are well in compliance with available experimental and theoretical values. The analysis of the computed elastic constants demonstrated that all three compounds are mechanically stable. According to Pugh's criteria of ductile or brittle, the studied materials have a ductile nature. The computed electronic band structure and their total and partial density of states are in a good agreement with existing data. Band structures and state densities confirmed the metallic character of  $X_3\text{Au}$  compounds in A15 structure. Phonon-dispersion curves and their corresponding projected density of states of  $X_3\text{Au}$  ( $X = \text{Ti, Zr and V}$ ) compounds in structure A15 were obtained using MedeA-Phonon in conjunction with MedeA-Vasp by the direct method.

#### References

- [1] V. Sathyakumar, S. Sankar, K. Mahalakshmi, Ab initio studies of thermal and superconducting properties of  $\text{Ti}_3\text{M}$  intermetallic compounds ( $\text{M} = \text{Au, Pt and Ir}$ ), *High Temp.* 44 (2) (2015).
- [2] Y. Chang, The float-zone growth of  $\text{Ti}_3\text{Au}$  and  $\text{Ti}_3\text{Pt}$ , *J. Crystal Growth* 62 (3) (1983) 627–632.
- [3] Y. Shinohara, et al., Deformation behavior of  $\text{Ti-4Au-5Cr-8Zr}$  superelastic alloy with or without containing  $\text{Ti}_3\text{Au}$  precipitates, *Materials Today: Proceed.* 2 (2015) S821–S824.
- [4] Y.-R. Lee, et al., Effect of gold addition on the microstructure, mechanical properties and corrosion behavior of Ti alloys, *Gold Bull.* 47 (3) (2014) 153–160.
- [5] M. Rajagopalan, R. Rajiv Gandhi, First principles study of structural, electronic, mechanical and thermal properties of A15 intermetallic compounds  $\text{Ti}_3\text{X}$  ( $\text{X} = \text{Au, Pt, Ir}$ ), *Physica. B* 407 (24) (2012) 4731–4734.
- [6] A.K. Kumar, A. Paul, Interdiffusion and activation energy in  $\text{Ti}_3\text{Au}$  phase with A15 crystal structure, *Intermetall.* 17 (11) (2009) 962–964.
- [7] S.V. Reddy, S.V. Suryanarayana, X-ray determination of the Debye-Waller factors, amplitudes of vibration and Debye temperatures of some intermetallic compounds with the A15-structure, *J. Mat. Sci. Lett.* 5 (4) (1986) 436–438.
- [8] S. Ramakrishnan, G. Chandra, Resistivity studies on low- $T_c$  A15 compounds, *Phy. Rev. B.* 38 (13) (1988) 9245–9247.
- [9] S. Ramakrishnan, A. Nigam, G. Chandra, Resistivity and magnetoresistance studies on superconducting A15  $\text{V}_3\text{Ga}$ ,  $\text{V}_3\text{Au}$ , and  $\text{V}_3\text{Pt}$  compounds, *Phy. Rev. B.* 34 (9) (1986) 6166.
- [10] A. Junod, R. Flukiger, J. Muller, Supraconductivite et chaleur specifique dans les alliages A15 a base de titane, *J. Phy. Chem. Solids* 37 (1) (1976) 27–31.
- [11] E. Svanidze, et al., High hardness in the biocompatible intermetallic compound  $\beta\text{-Ti}_3\text{Au}$ , *Sci. Adv.* 2 (7) (2016) e1600319.
- [12] Z.-h. Li, Z.-p. Jin, H.-s. Liu, Thermodynamic assessment of Au-Zr system, *J. Cent. South Univ. Tech.* 10 (2) (2003) 94–97.
- [13] M. Lomello-Tafin, et al., The Zr-Au system, *J. Alloys Comp.* 257 (1) (1997) 215–223.
- [14] B.T. Matthias, T.H. Geballe, V.B. Compton, Superconductivity, *Rev. Mod. Phy.* 35 (1) (1963) 1–22.
- [15] E. Stolz, K. Schubert, Structure investigations on some systems homologous and quasi-homologous to  $4\text{ T} - 1\text{ B}$ , *Z. Metallkd.* 53 (7) (1962) 433–444.

- [16] M. Nevitt, Atomic size effects in Cr<sub>3</sub>O-type structure, *Trans. AIME* 212 (1958) 350–355.
- [17] W.B. Pearson, Dimensional analysis of phases with the A15 β-W structure, *J. Less Common Metals* 77 (2) (1981) 227–240.
- [18] D.I. Bardos, et al., The magnetic susceptibility of transition-metal A15-type phases, *J. Low Temp. Phys.* 3 (5) (1970) 509–518.
- [19] E. Van Reuth, R. Waterstrat, Atomic ordering in binary A15-type phases, *Acta Crystallographica Section B: Structural Crystallography and Crystal Chemistry* 24 (2) (1968) 186–196.
- [20] S. Geller, A set of effective coordination number (12) radii for the β-wolfram structure elements, *Acta Crystallographica* 9 (11) (1956) 885–889.
- [21] M. Wang, et al., Structural, elastic and thermodynamic properties of A15-type compounds V<sub>3</sub>X (X= Ir, Pt and Au) from first-principles calculations, *Mod. Phys. Lett. B.* (2016) 1650414.
- [22] R. Blaugher, et al., Atomic ordering and superconductivity in A-15 compounds, *J. Low Temp. Phys.* 1 (6) (1969) 539–561.
- [23] A. Van Winkel, et al., Diffusion of 48V and 67Ga in the A15 intermetallic compound V<sub>3</sub>Ga, *J. Less Common Metals* 99 (2) (1984) 257–264.
- [24] A. Kumar, et al., Determination of diffusion parameters and activation energy of diffusion in V<sub>3</sub>Si phase with A15 crystal structure, *Scripta Materialia* 60 (6) (2009) 377–380.
- [25] P. Spitzli, et al., Chaleur spécifique, supraconductivité et ordre cristallin d'alliages de structure A15 à base de vanadium, *J. Phys. Chem. Solids* 31 (7) (1970) 1531–1537.
- [26] T. Jarlborg, A. Junod, M. Peter, Electronic structure, superconductivity, and spin fluctuations in the A 15 compounds A<sub>3</sub>B: A = V, Nb; B = Ir, Pt, Au, *Physic. Rev. B.* 27 (3) (1983) 1558.
- [27] T. Smith, Superconductivity at high pressure in the A15 compounds, *J. Low Temp. Phys.* 6 (1) (1972) 171–195.
- [28] G. Kresse, J. Furthmüller, Efficient iterative schemes for ab initio total-energy calculations using a plane-wave basis set, *Physic. Rev. B.* 54 (16) (1996) 11169–11186.
- [29] G. Kresse, J. Hafner, Ab initio molecular-dynamics for liquid-metals, *Physic. Rev. B.* 47 (1) (1993) 558–561.
- [30] P.E. Blochl, Projector augmented-wave method, *Physic. Rev. B.* 50 (24) (1994) 17953–17979.
- [31] J.P. Perdew, K. Burke, M. Ernzerhof, Generalized gradient approximation made simple, *Physic. Rev. Lett.* 77 (18) (1996) 3865–3868.
- [32] H.J. Monkhorst, J.D. Pack, Special points for Brillouin-zone integrations, *Physic. Rev. B.* 13 (12) (1976) 5188–5192.
- [33] M. Methfessel, A.T. Paxton, High-precision sampling for Brillouin-zone integration in metals, *Physic. Rev. B.* 40 (6) (1989) 3616–3621.
- [34] Y. Le Page, P. Saxe, Symmetry-general least-squares extraction of elastic coefficients from ab initio total energy calculations, *Physic. Rev. B.* 63 (17) (2001).
- [35] M. Born, K. Huang, *Dynamical Theory of Crystal Lattices*, Clarendon press, 1954.
- [36] S.F. Pugh, XCII. Relations between the elastic moduli and the plastic properties of polycrystalline pure metals, *Lon. Edin. Dub. Phil. Mag. J. Sci.* 45 (367) (1954) 823–843.
- [37] N.F. Mott, H. Jones, *The Theory of the Properties of Metals and Alloys*, Courier Corporation, 1958.
- [38] N. Arıkan, et al., Structural, elastic, electronic and phonon properties of scandium-based compounds ScX<sub>3</sub> (X = Ir, Pd, Pt and Rh): An ab initio study, *Comput. Mat. Sci.* 79 (2013) 703–709.
- [39] N. Arıkan, et al., Electronic and phonon properties of the full-Heusler alloys X<sub>2</sub>YAl (X = Co, Fe and Y = Cr, Sc): a density functional theory study, *J. Mat. Sci.* 49 (12) (2014) 4180–4190.